

THz generation at 1.55 μm excitation: six-fold increase in THz conversion efficiency by separated photoconductive and trapping regions

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Abstract: We present first results on photoconductive THz emitters for 1.55 μm excitation. The emitters are based on MBE grown $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}/\text{In}_{0.52}\text{Al}_{0.48}\text{As}$ multilayer heterostructures (MLHS) with high carrier mobility. The high mobility is achieved by spatial separation of photoconductive and trapping regions. Photoconductive antennas made of these MLHS are evaluated as THz emitters in a THz time domain spectrometer (THz TDS). The high carrier mobility and effective absorption significantly increases the optical-to-THz conversion efficiency with THz bandwidth in excess of 3 THz.

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1. Introduction

Terahertz time domain spectroscopy (TDS) is by far the method of most importance within the rapidly developing and prosperous field of terahertz technology [1]. This method is based on the optical or optoelectronic generation and detection of short THz pulses by non-linear crystals or semiconductor based photoconductive antennas (PCA). In the latter case it is crucial to employ a material that exhibits very short carrier lifetimes in order to obtain broadband THz spectra.

The first generation of THz radiation from PCAs was achieved with radiation damaged silicon-on-sapphire [2,3]. Later experiments employed low temperature (LT) molecular beam epitaxy (MBE) grown GaAs, which rapidly became the state of the art material [4–6]. For all material systems the THz pulses were excited by titanium sapphire femtosecond lasers at wavelengths around 800 nm.

In the last 20 years there have been numerous approaches to increase the performance of photoconductive antennas, e.g. investigating antenna structure [7,8], metallization [9] and carrier lifetime [10] or by utilizing new material systems, such as GaAs:ErAs [11,12], LT GaAsSb [13] and GaAsBi [14]. The first attempt of PCAs based on InGaAs, for excitation with cost-effective fibre lasers at wavelengths around 1.55 μm , was made by Suzuki and Tonouchi [15,16] with ion-implanted InGaAs, followed by Takazato and Kadoya with LT-growth of InGaAs [17,18]. Subsequently, it was shown that LT MBE grown Beryllium doped InGaAs/InAlAs heterostructures are suitable for broadband THz emitters and detectors [19]. Followed by the introduction of a completely fibre coupled THz TDS system based on these antennas [20]. Since then other groups also demonstrated THz emission and detection in InGaAs based materials [21–23].

Despite these great efforts there is still vast potential for improvement of InGaAs based PCAs. As mentioned above, short carrier lifetimes are a key feature for broadband THz PCAs. These short carrier lifetimes are generally realized by inducing defect states into the respective semiconductor material. These defect states can be realized by strong doping [22,23], ion implantation [15,16], growth conditions, e.g. LT growth [17–19], or the growth of special recombination centers [21]. Additional important characteristics for efficient THz emitters are efficient absorption, a sufficiently high carrier mobility and high dark resistivity, i.e. low residual carrier concentrations.

However, it is difficult to fulfill all of the above requirements, since high defect material typically shows a strongly reduced carrier mobility due to elastic and inelastic (i.e. trapping) scattering of carriers at defect sites. Furthermore, in case of InGaAs the defects states are energetically situated relatively close to the conduction band. This shifts the Fermi level closer to the conduction band edge which results in low dark resistivity at room temperature.

The Fermi level can be lowered by counter doping with an acceptor-type dopant. However, counter doping further reduces carrier mobility. In addition, light absorption is also reduced in high defect materials.

In this work we present a new approach to circumvent some of these obstacles. The basic idea is to spatially separate the photoconductive region, i.e. where light absorption and carrier transport take place, from regions that exhibit high defect densities and that are transparent for 1.55 μm excitation, thus solely acting as trapping and recombination regions.

2. Principle and growth

A device meeting the above mentioned requirements can be realized by MBE growth of $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}/\text{In}_{0.52}\text{Al}_{0.48}\text{As}$ multi-layer heterostructures (MLHS) (as depicted in Fig. 1(a)) when utilizing a special characteristic of MBE growth of InAlAs. Within a substrate temperature range between $T_s = 300 - 500$ °C the growth of InAlAs shows strong alloy clustering effects with InAs-like and AlAs-like regions featuring cluster sizes of several nanometers, with a maximum cluster density for $T_s \approx 400$ °C [24]. The activation energies of these cluster defects have been measured to be in the region of $E_A = 0.6-0.7$ eV [24,25] which results in the InAlAs to be semi-insulating. For the growth of a MLHS at this temperature and considering the conduction band offset between InAlAs ($E_g = 1.47$ eV at 300 K) and InGaAs ($E_g = 0.74$ eV at 300 K) of approximately $\Delta E_c = 0.44$ eV [26], this results in defect states within the InAlAs layers that are energetically situated significantly below the conduction band of adjacent InGaAs layers, as depicted schematically in Fig. 1(b).

If the InAlAs layers are sufficiently thin, i.e. the the electron wave function of an optically excited electron in the conduction band of an InGaAs layer has a sufficiently large overlap with the cluster defect states in the adjacent InAlAs layer, these defects can act as effective traps for those electrons. In addition, the MBE growth of InGaAs at substrate temperatures

around 400 - 500 °C shows a minimum for the residual carrier concentration, with $N_A - N_D < 3 \cdot 10^{-15} \text{ cm}^{-3}$ and Hall mobility values for bulk material of $\mu_{H, \text{InGaAs}} = 10000 \text{ cm}^2/\text{Vs}$ [27]. Hence, the growth of a InGaAs/InAlAs MLHS should lead to short carrier lifetimes while maintaining effective absorption, high dark resistivity and high carrier mobility in the photoconductive layer.

The InGaAs/InAlAs MLHS investigated in this work were grown by elemental source molecular beam epitaxy on semi insulating InP:Fe substrates at an approximate substrate temperature of 400 °C. First a 777 nm InAlAs buffer layer was grown followed by 100 periods of 12 nm InGaAs layers and 8 nm InAlAs layers.

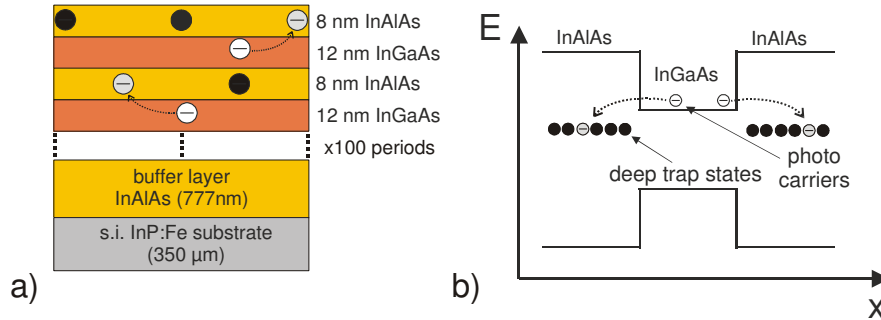


Fig. 1. (a) Schematic of InGaAs/InAlAs heterostructure, with 100 periods of a 12 nm InGaAs layer followed by a 8 nm InAlAs layer with cluster-induced defects acting as electron traps. 1(b) Schematic of the respective band-diagram in real space with deep cluster-induced defect states.

We measured Hall mobility values for the MLHS grown at $T_s = 400 \text{ °C}$ of $\mu_{H,400\text{-MLHS}} = 1500\text{-}3000 \text{ cm}^2/\text{Vs}$, with sheet carrier densities down to $N_s = 2.3 \cdot 10^8 \text{ cm}^{-2}$ and resistivity values of up to $\rho = 5800 \text{ } \Omega \cdot \text{cm}$. The decrease in Hall mobility compared to the bulk InGaAs value is due to the fact that trapping of carriers into defect states in the InAlAs layers also contributes to the scattering time which is probed by hall mobility measurements. Nevertheless the mobility of the MLHS grown at 400°C is still almost one magnitude higher than that of LT-grown ($T_s = 130\text{°C}$) Be-doped MLHS with Hall mobility values of $\mu_{H, \text{LT-MLHS}} < 500 \text{ cm}^2/\text{Vs}$, with $N_s = 1.4 \cdot 10^{10} \text{ cm}^{-2}$ and $\rho = 240 \text{ } \Omega \cdot \text{cm}$.

3. THz TDS measurements

In order to evaluate the 400 °C grown material as a THz emitter, the samples were processed as mesa-type antennas as described in [28] with a stripline type antenna geometry and a stripline separation of 25 μm. A similarly structured conventional LT grown Be-doped InGaAs/InAlAs MLHS served as a reference emitter in all measurements. For detection we used a photoconductive receiver also based on LT-grown Be-doped InGaAs/InAlAs MLHS processed with a 10 μm gap dipole mesa-type antenna. The THz TDS setup consisted of a pre-compensated pulsed Er-doped fibre laser with a repetition rate of 100 MHz and pulses with approximately 80 fs FWHM pulse width. The laser was focused onto the emitter and detector with a spot size of approximately 10 μm. The THz beam path consisted of hyper-hemispherical silicon lenses attached to the backside of each antenna and two off-axis parabolic mirrors in between to focus the THz emission onto the detector. The overall THz path length was approximately 20 cm. A mechanical delay stage was used to introduce a delay between the optical pump and probe pulse, with a used time resolution of 50 fs. The emitters were fed with a bias modulated from 0 V to 5 V and a modulation frequency of 500 Hz. The lock-in time constant was 30 ms. Figure 2 shows the Fourier spectrum of the signal from the LT-reference antenna with the corresponding THz pulse trace in the inset. Figure 3 shows the FFT spectrum and THz pulse obtained from the new emitter material. The optical excitation was 10 mW at the emitter and 20 mW at the detector.

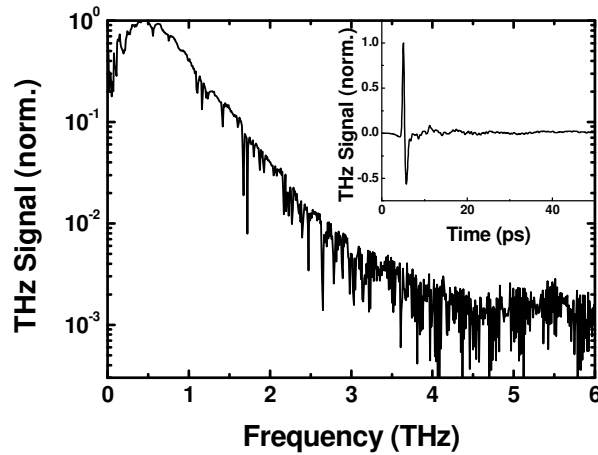


Fig. 2. THz pulse trace and corresponding FFT spectrum for a conventional LT-grown Be-doped MLHS THz emitter grown at $T_s = 130$ °C (this is serves as a reference).

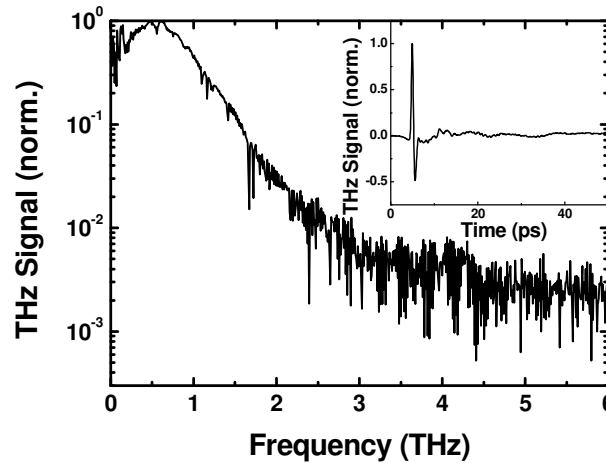


Fig. 3. THz pulse trace and corresponding FFT Spectrum for a MLHS grown at $T_s = 400$ °C, with separated trapping and photoconductive regions.

As can be seen the spectral bandwidth of the new design is comparable to the one obtained with the LT grown reference, both extending well beyond 3 THz. The high bandwidth obtained with the new device suggests that the device exhibits relatively fast carrier trapping times, thus supporting the assumption of an effective trapping mechanism within the device. To further assess the carrier life time a dipole antenna, similar to that of the before used LT-grown detector, was deposited on the 400 °C grown material. This detector was then used in the TDS setup employing the LT-grown reference emitter as a source. The obtained FFT spectrum and pulse trace are depicted in Fig. 4. The detected THz spectrum extends beyond 3 THz, which further supports the assumption of short carrier life times in the 400 °C grown MLHS.

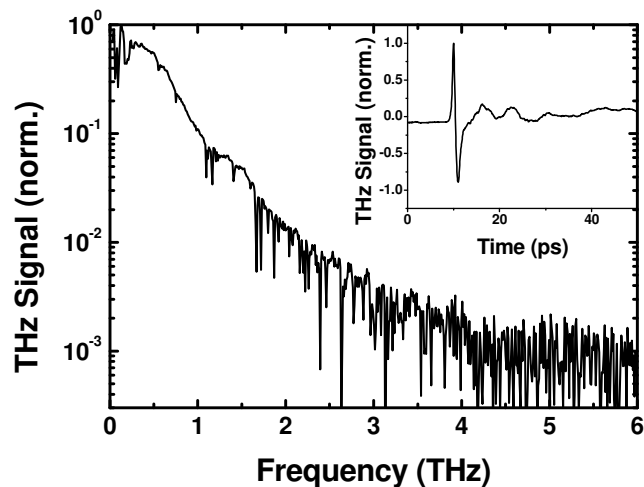


Fig. 4. THz pulse trace and corresponding FFT spectrum for the LT-grown Be-doped MLHS reference emitter grown at $T_s = 130^\circ\text{C}$ and a $T_s = 400^\circ\text{C}$ grown receiver with a Dipol antenna.

4. Dependence on bias field and optical-to-THz conversion efficiency

To further evaluate the new material, we measured the peak-to-peak amplitude of the THz pulse obtained by TDS measurements in dependence of the applied bias field as well as in dependence on the optical power incident on the emitter antenna. Again an LT grown Be-doped MLHS antenna served as a reference. The THz amplitude over applied bias field is shown in Fig. 4. As can be seen the new MLHS grown at 400°C shows a much stronger THz emission than the LT reference. We attribute this to the improved carrier mobility in the new material. This can be understood within the framework of the classical Maxwell and Drude-model [29], where the emitted THz field is directly proportional to the derivative of the time varying photocurrent which is proportional to mobility. The measured DC photocurrents in the 400°C MLHS were strongly increased, e.g. for a bias field of 2 kV/cm the DC photocurrents were $I_p = 516\ \mu\text{A}$ for the MLHS grown at 400°C and $I_p = 9\ \mu\text{A}$ for the LT grown material. This significant difference can't be explained by the higher mobility alone. We assume that the trapping time for the 400°C grown MLHS is slightly higher than for LT-grown MLHS. Additionally, the density of trap states in the 400°C is assumed to be smaller than for the LT-grown MLHS reference. This in turn results in a higher amount of long-lived carriers. This fact combined with the higher mobility lead to a strongly increased DC photocurrent. In Fig. 5 the THz amplitude is shown in dependence of the optical excitation power at the emitter for a constant bias field of 2 kV/cm . The new material shows a significantly higher light sensitivity compared to the LT reference, which we attribute to a shaper absorption band edge and the higher mobility in the 400°C grown material compared to the LT-grown material.

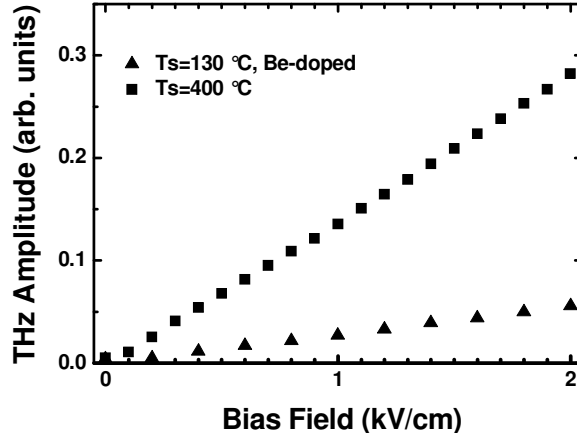


Fig. 5. Emitted THz-pulse amplitude detected by a PCA receiver in a THz TDS setup, as a function of applied bias field at the emitter for a MLHS grown at $T_s = 400\text{ }^\circ\text{C}$ (squares) and a MLHS grown at $T_s = 130\text{ }^\circ\text{C}$, Be-doped (triangles). The applied optical power was 10 mW for both emitter and receiver.

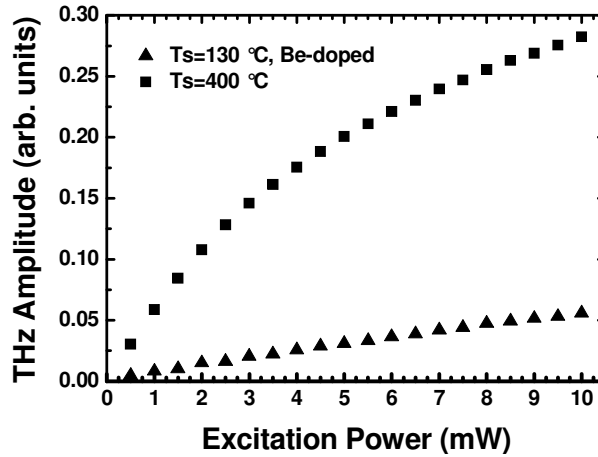


Fig. 6. Emitted THz-pulse amplitude detected by a PCA receiver in a THz TDS setup, as a function of optical excitation power at the emitter for a MLHS grown at $T_s = 400\text{ }^\circ\text{C}$ (squares) and a MLHS grown at $T_s = 130\text{ }^\circ\text{C}$, Be-doped (triangles). The applied emitter bias field was 2 kV/cm.

5. Conclusion and outlook

We presented a new concept and the first realization of InGaAs based THz emitters that combine fast trapping times with high mobility and efficient absorption. The new emitters are capable of broadband THz emission while raising the optical-to-THz conversion efficiency by almost one order of magnitude. Future designs comprising optimized growth conditions and trapping layer thicknesses will potentially further improve the performance of these pulsed THz emitters.

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